

# **Radiation Tolerant Detectors for Future HEP Experiments**

**Results from the  
CERN-RD50 Collaboration**

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University of Hamburg

See also:

Michael Moll

**Radiation Tolerant Sensors for Pixel Detectors**

- CERN-RD50 project -

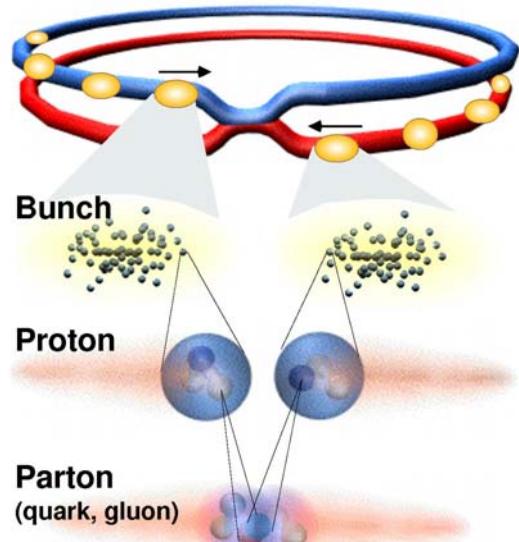
Pixel2005 Bonn September 05

**<http://www.cern.ch/rd50>**

# Outline

- ❑ Silicon Detectors for Particle Tracking
- ❑ Motivation for R&D, the Challenge for Radiation Tolerance
- ❑ The RD50 Collaboration
- ❑ Radiation Damage, Deterioration of Detector Properties
- ❑ Approaches for Solutions, Material and Device Engineering
- ❑ Summary and Outlook

# Silicon Detectors: Favorite Choice for Particle Tracking



## LHC properties

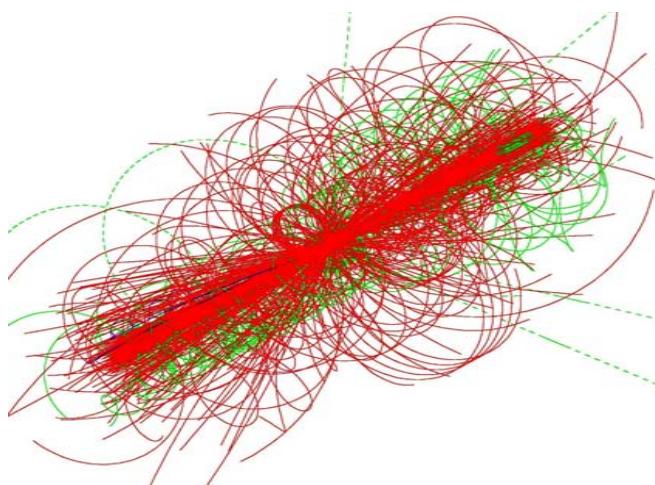
Proton-proton collider, 2 x 7 TeV

Luminosity:  $10^{34}$

Bunch crossing: every 25 nsec, Rate: 40 MHz  
event rate:  $10^9/\text{sec}$  (23 interactions per bunch crossing)

Annual operational period:  $10^7$  sec

Expected total op. period: 10 years



### Experimental requests

Reliable detection of mips →  $\text{S/N} \approx 10$

High event rate → time + position resolution:  
high track accuracy →  $\sim 10 \text{ ns}$  and  $\sim 10 \mu\text{m}$

Complex detector design → low voltage operation in  
normal ambients

Intense radiation field during 10 years → Radiation tolerance up to  
 $10^{15} \text{ hadrons/cm}^2$

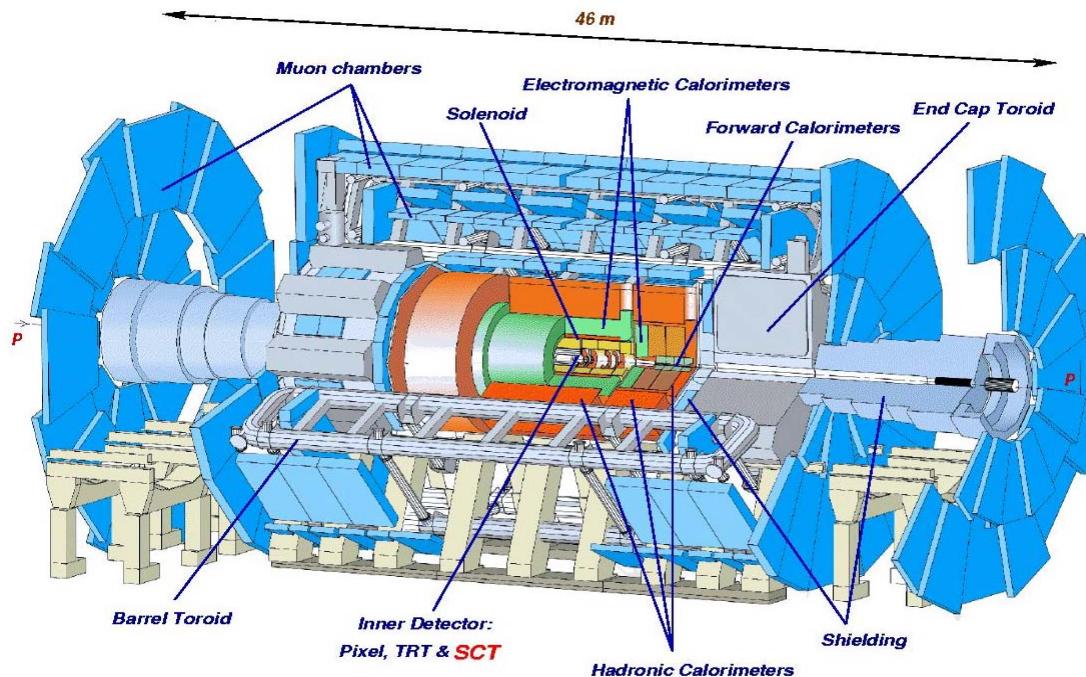
Feasibility, e.g. 200 m<sup>2</sup> for CMS → large scale availability  
known technology, low cost

### Detector properties

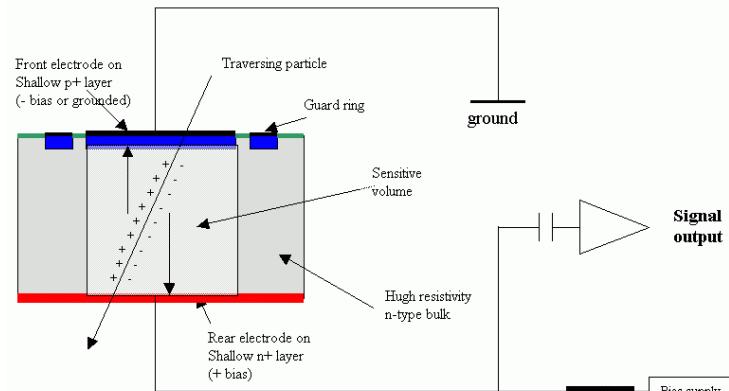
! Silicon Detectors meet all Requirements !

# LHC ATLAS Detector – a Future HEP Experiment

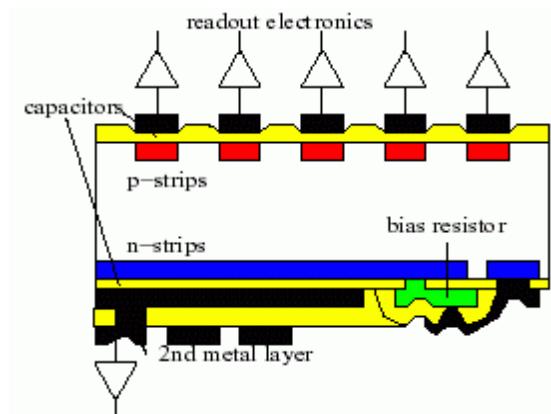
Overall length: 46m, diameter: 22m,  
total weight: 7000t, magnetic field: 2T  
ATLAS collaboration: 1500 members



*principle of a silicon detector:  
solid state ionization chamber*



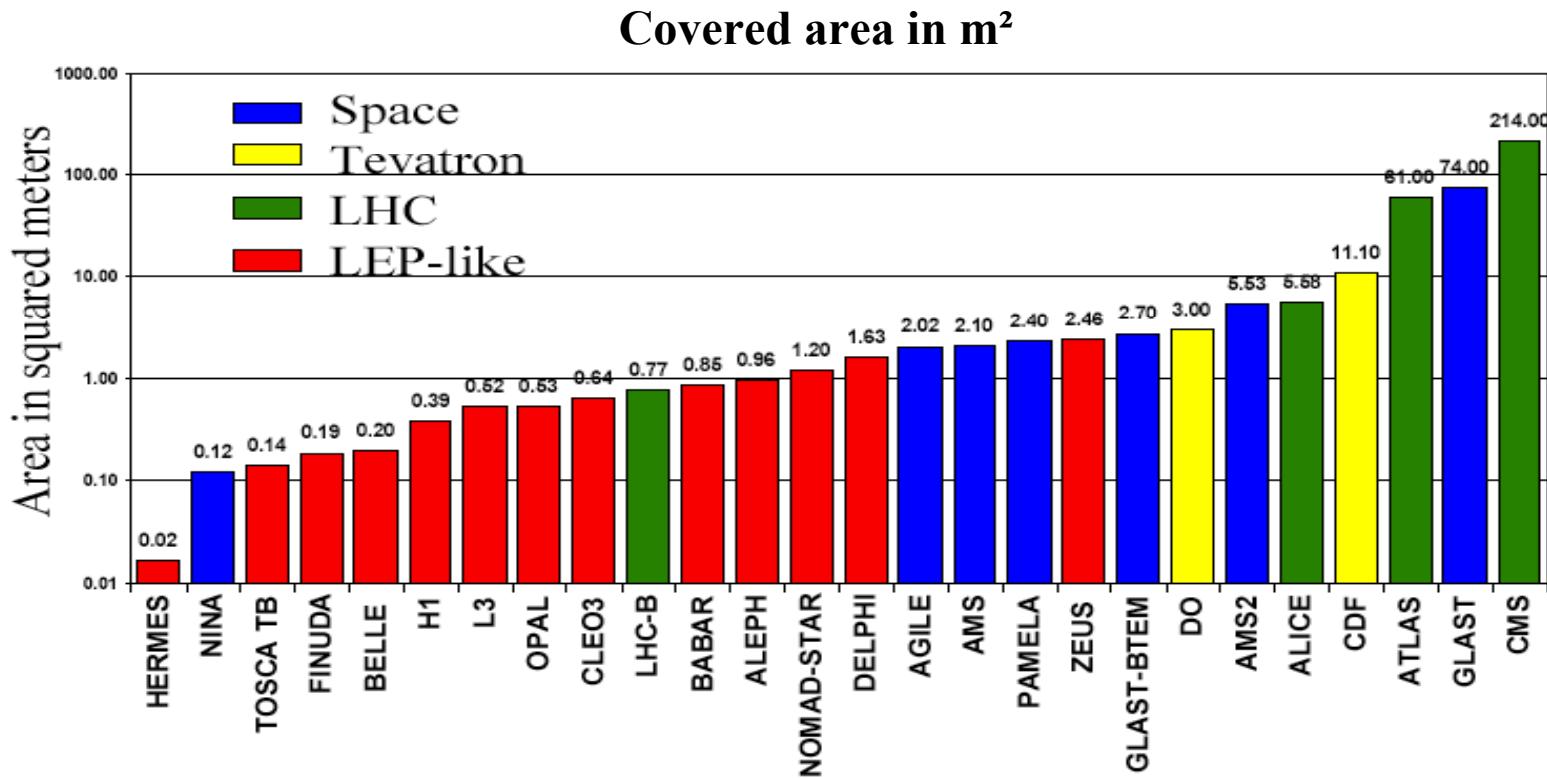
*micro-strip detector  
for particle tracking*



# Growing demand for Si-detectors in tracking applications



## Experiments using silicon strip detectors



E. do Couto e Silva – SLAC/Stanford University

Vertex 2000, Sept 10-15, National Lakeshore, MI, USA

# Main motivations for R&D on Radiation Tolerant Detectors: Super - LHC

- **LHC upgrade**

⇒ **LHC (2007)**,  $L = 10^{34} \text{cm}^{-2}\text{s}^{-1}$

**10 years**  
→ **500 fb<sup>-1</sup>**

$$\phi(r=4\text{cm}) \sim 3 \cdot 10^{15} \text{cm}^{-2}$$

**CERN-RD48**

⇒ **Super-LHC (2015 ?)**,  $L = 10^{35} \text{cm}^{-2}\text{s}^{-1}$

**5 years**  
→ **2500 fb<sup>-1</sup>**

$$\phi(r=4\text{cm}) \sim 1.6 \cdot 10^{16} \text{cm}^{-2}$$

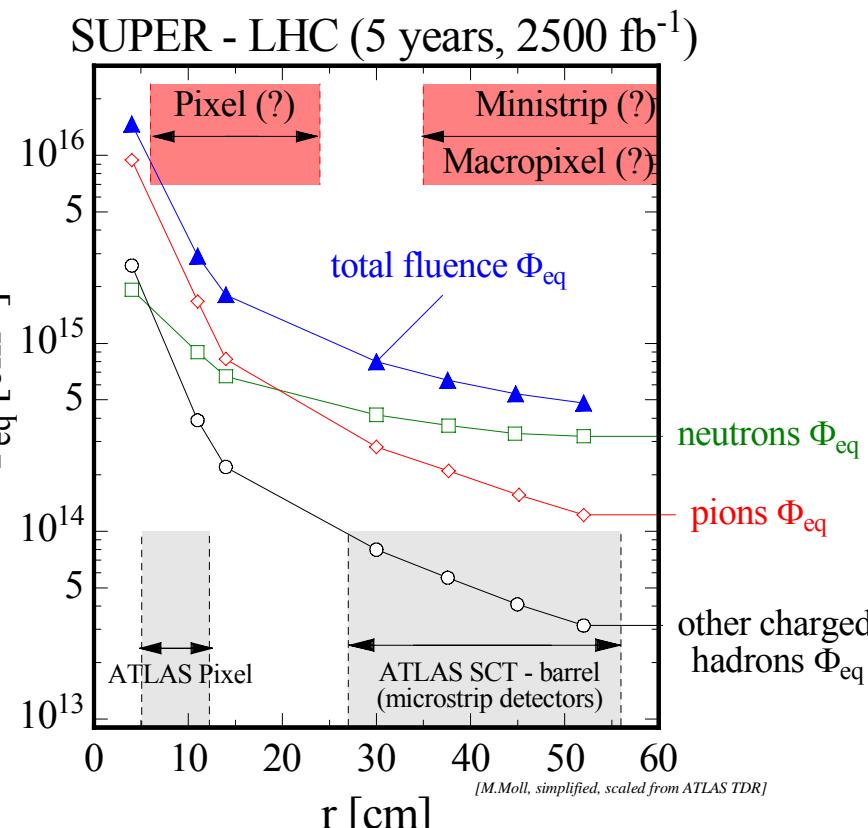
**CERN-RD50**

- **LHC (Replacement of components)**

e.g. - LHCb Velo detectors (~2010)  
- ATLAS Pixel B-layer (~2012)

- **Linear collider experiments (generic R&D)**

Deep understanding of radiation damage will be fruitful for linear collider experiments where high doses of  $e, \gamma$  will play a significant role.



Adopted from M. Moll, CERN, Bonn, Sep-05

RD50: Development of Radiation Hard Semiconductor Devices for High Luminosity Colliders

- Collaboration formed in November 2001
- Experiment approved as RD50 by CERN in June 2002
- Main objective:

**Development of ultra-radiation hard semiconductor detectors for the luminosity upgrade of the LHC to  $10^{35} \text{ cm}^{-2}\text{s}^{-1}$  (“Super-LHC”).**

**Challenges:**

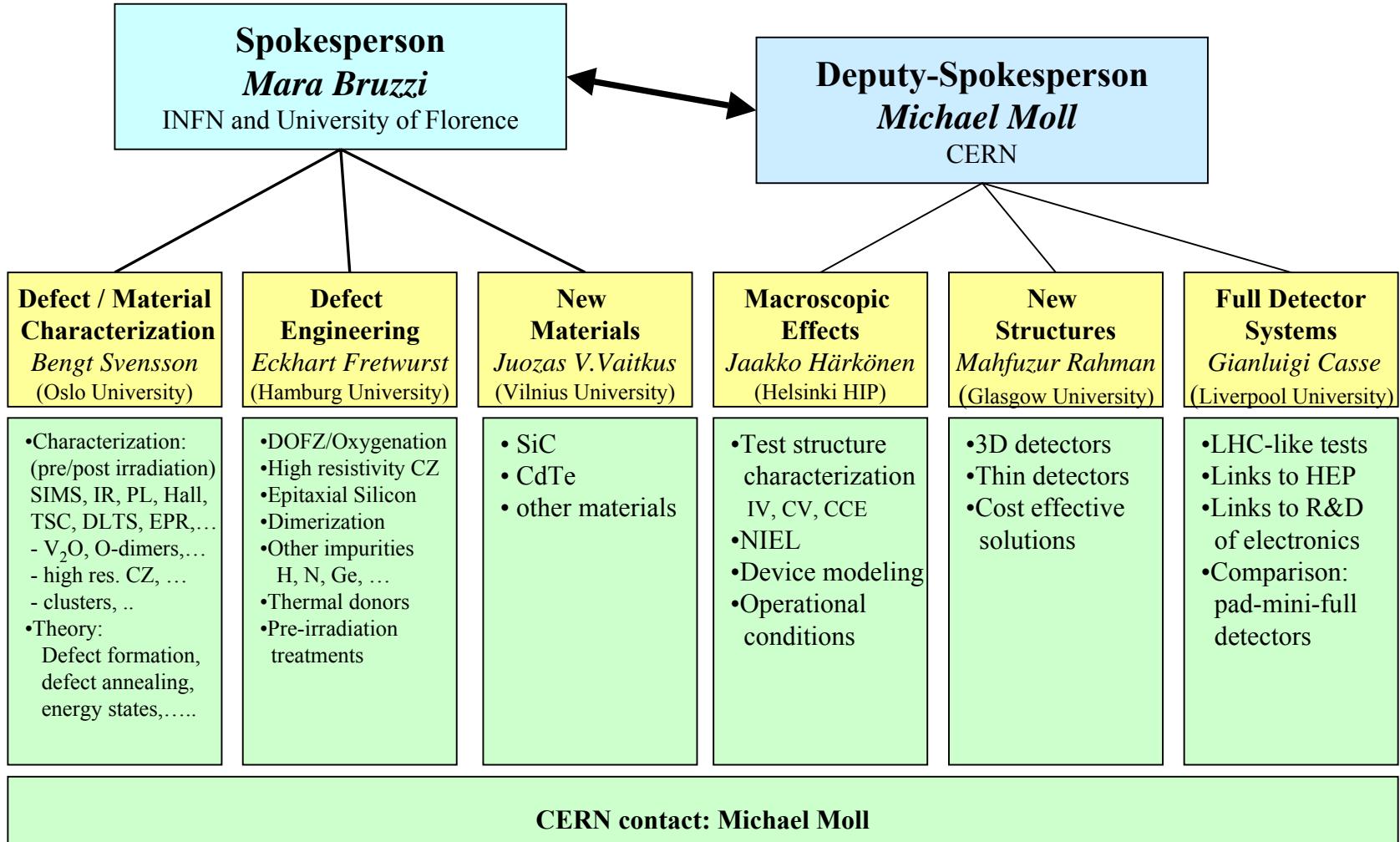
- Radiation hardness up to  $10^{16} \text{ cm}^{-2}$  required
- Fast signal collection (Going from 25 ns to 10 ns bunch crossing ?)
- Low mass (reducing multiple scattering close to interaction point)
- Cost effectiveness (big surfaces have to be covered with detectors!)

## Presently 251 members from 51 institutes

Belarus (Minsk), Belgium (Louvain), Canada (Montreal), Czech Republic (Prague (3x)), Finland (Helsinki, Lappeenranta), Germany (Berlin, Dortmund, Erfurt, Freiburg, Hamburg, Karlsruhe), Israel (Tel Aviv), Italy (Bari, Bologna, Florence, Padova, Perugia, Pisa, Trento, Turin), Lithuania (Vilnius), Norway (Oslo (2x)), Poland (Warsaw(2x)), Romania (Bucharest (2x)), Russia (Moscow), St.Petersburg), Slovenia (Ljubljana), Spain (Barcelona, Valencia), Switzerland (CERN, PSI), Ukraine (Kiev), United Kingdom (Exeter, Glasgow, Lancaster, Liverpool, Sheffield, University of Surrey), USA (Fermilab, Purdue University, Rochester University, SCIPP Santa Cruz, Syracuse University, BNL, University of New Mexico)

*Adopted from M. Moll, CERN, Bonn, Sep-05*

# Scientific Organization of RD50



# Radiation Damage in Silicon Sensors

## □ Two general types of radiation damage in detector materials:

- Bulk (Crystal) damage due to Non Ionizing Energy Loss (NIEL)

- displacement damage, built up of crystal defects –

- I. Change of effective doping concentration (higher depletion voltage, under-depletion)

- II. Increase of leakage current (increase of shot noise, thermal runaway)

- III. Increase of charge carrier trapping (loss of charge)

- Surface damage due to Ionizing Energy Loss (IEL)

- accumulation of positive in the oxide ( $\text{SiO}_2$ ) and the  $\text{Si}/\text{SiO}_2$  interface – affects: interstrip capacitance (noise factor), breakdown behavior, ...

## □ Impact on detector performance and Charge Collection

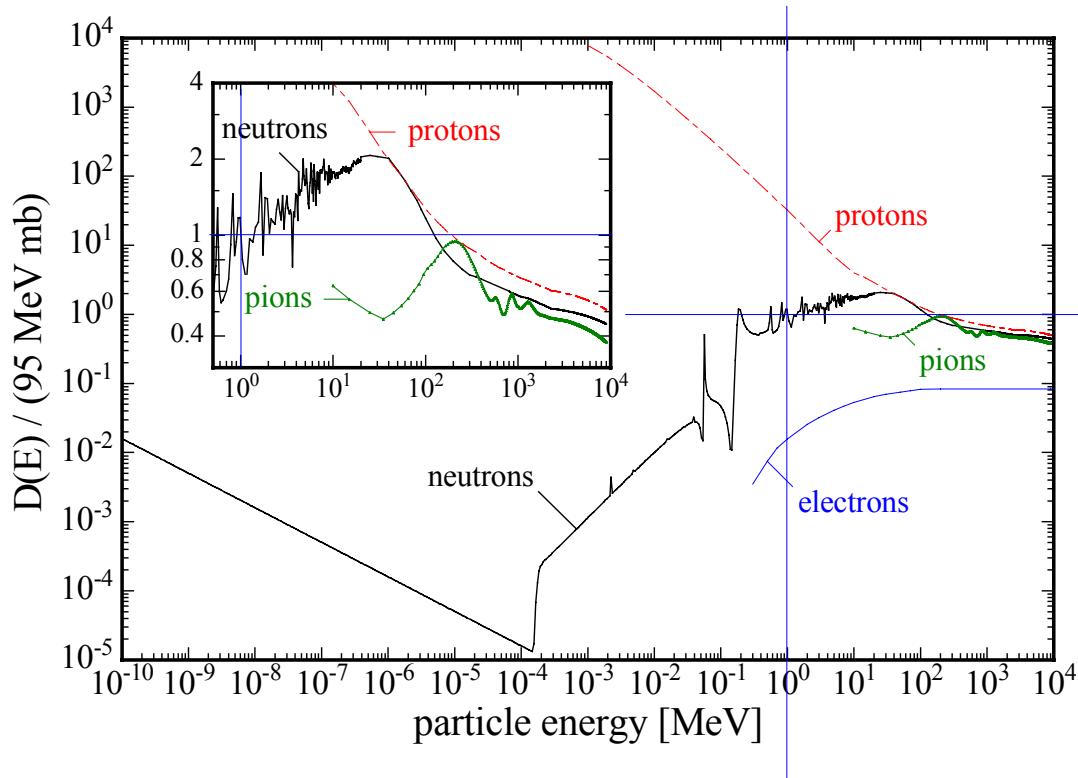
(depending on detector type and geometry and readout electronics!)

Signal/noise ratio is the quantity to watch

⇒ Sensors can fail from radiation damage !

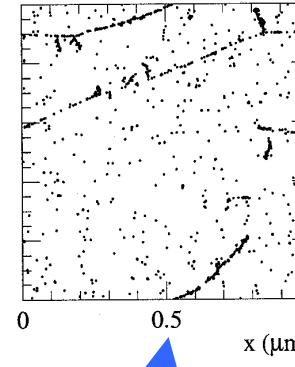
Adopted from M. Moll, CERN, Bonn, Sep-05

# Non Ionizing Energy Loss NIEL: displacement damage



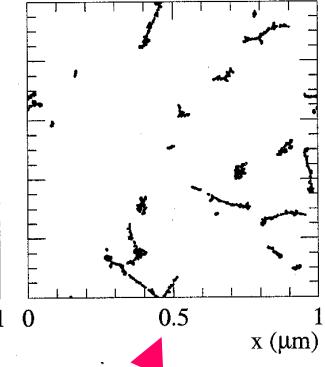
Point defects  
+ clusters

4145 vacancies



Dominated by  
clusters

8870 vacancies

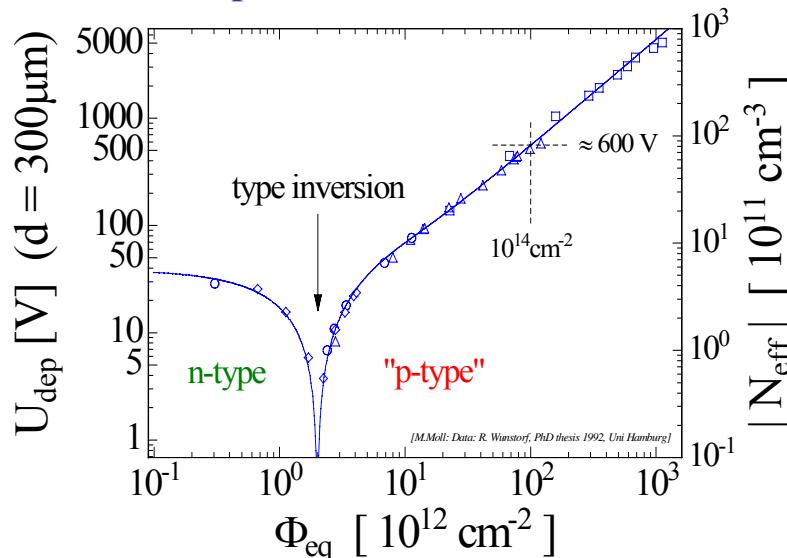


Damage effects generally  $\sim$  NIEL, however differences between proton & neutron damage

# Radiation Damage I. – Effective doping concentration

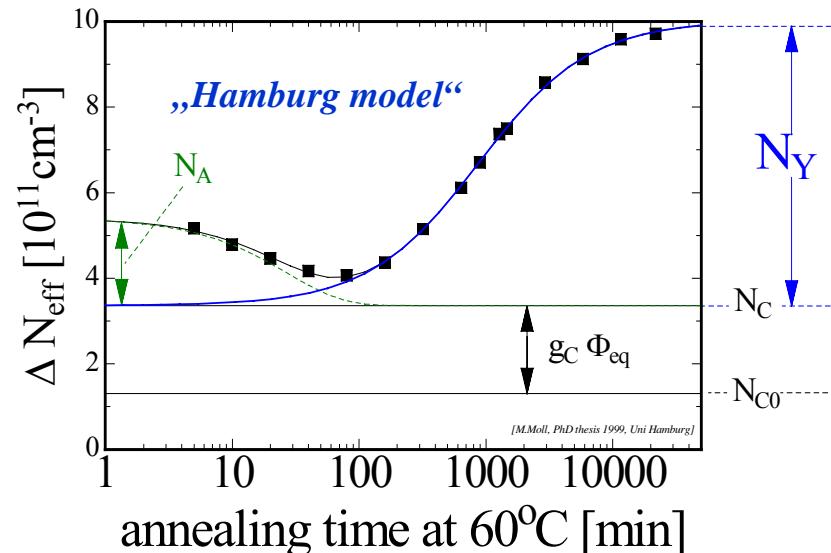
## Change of Depletion Voltage $V_{\text{dep}}$ ( $N_{\text{eff}}$ )

.... with particle fluence:



“Type inversion”:  $N_{\text{eff}}$  changes from positive to negative (Space Charge Sign Inversion)

.... with time (annealing):



Short term: “Beneficial annealing”

Long term: “Reverse annealing”

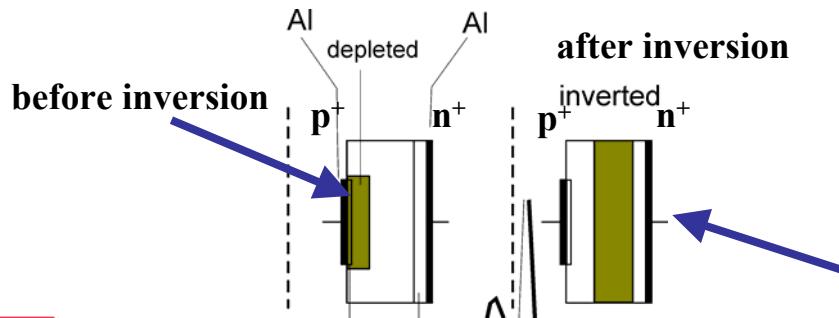
- time constant depends on temperature:

~ 500 years (-10°C)

~ 500 days ( 20°C)

~ 21 hours ( 60°C)

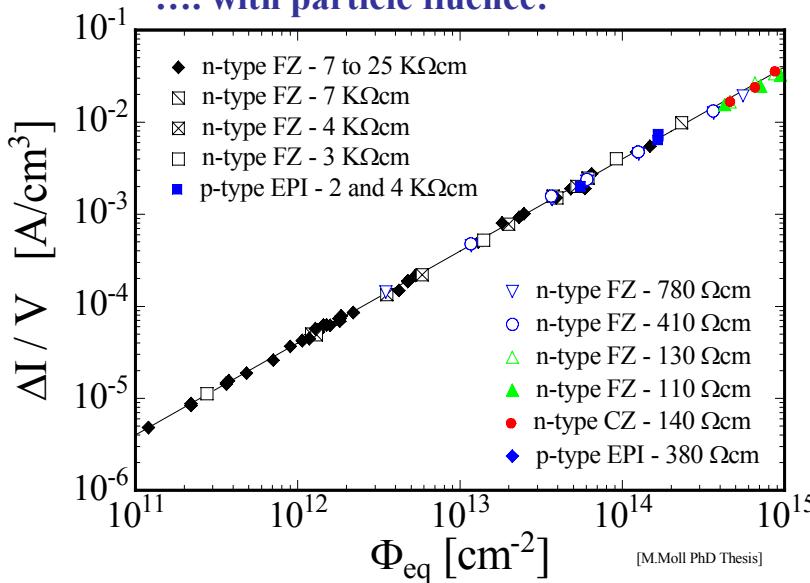
- Consequence: Detectors must be cooled even when the experiment is not running!



# Radiation Damage II. – Leakage current

- Change of Leakage Current (after hadron irradiation)

.... with particle fluence:



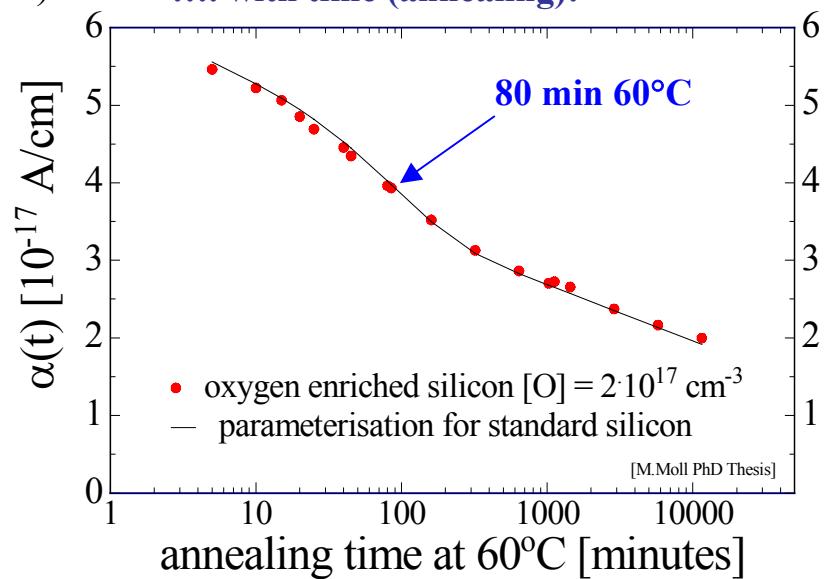
- Damage parameter  $\alpha$  (slope in figure)

$$\alpha = \frac{\Delta I}{V \cdot \Phi_{eq}}$$

**Leakage current  
per unit volume  
and particle fluence**

- $\alpha$  is constant over several orders of fluence and independent of impurity concentration in Si
  - ➡ can be used for fluence measurement

.... with time (annealing):



- Leakage current decreasing in time (depending on temperature)
- Strong temperature dependence:

$$I \propto \exp\left(-\frac{E_g}{2k_B T}\right)$$

**Consequence:**

Cool detectors during operation!  
Example:  $I(-10^\circ\text{C}) \sim 1/16 I(20^\circ\text{C})$

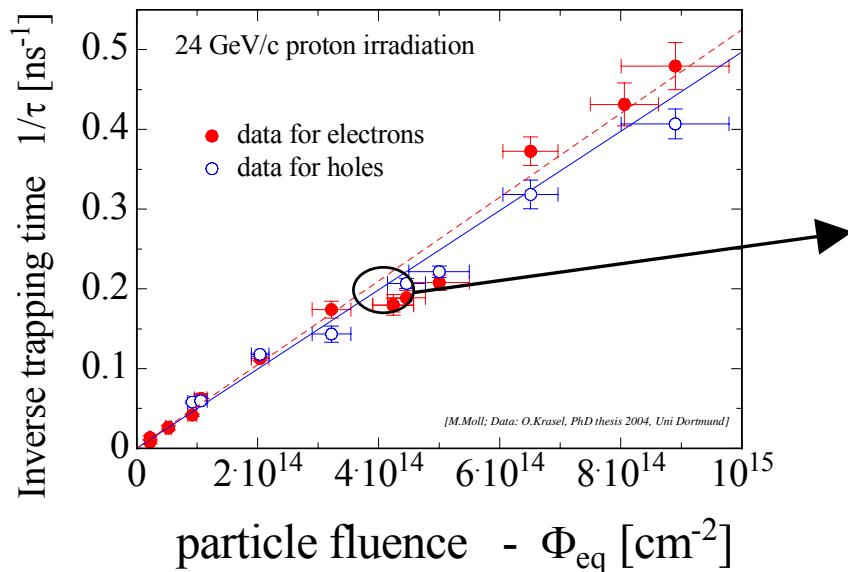
# Radiation Damage III. – Charge carrier trapping

## Deterioration of Charge Collection Efficiency (CCE) by trapping

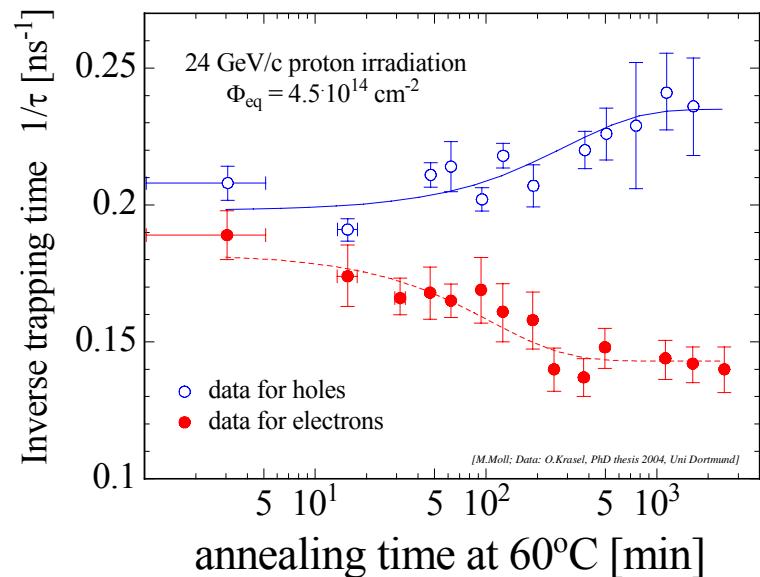
Trapping is characterized by an effective trapping time  $\tau_{\text{eff}}$  for electrons and holes:

$$Q_{e,h}(t) = Q_{0e,h} \exp\left(-\frac{1}{\tau_{\text{eff},e,h}} \cdot t\right) \quad \text{where:} \quad \frac{1}{\tau_{\text{eff},e,h}} \propto N_{\text{defects}}$$

### Increase of inverse trapping time ( $1/\tau$ ) with fluence



### ..... and change with time (annealing):



Adopted from M. Moll, CERN, Bonn, Sep-05

# Impact on Detector: Decrease of CCE

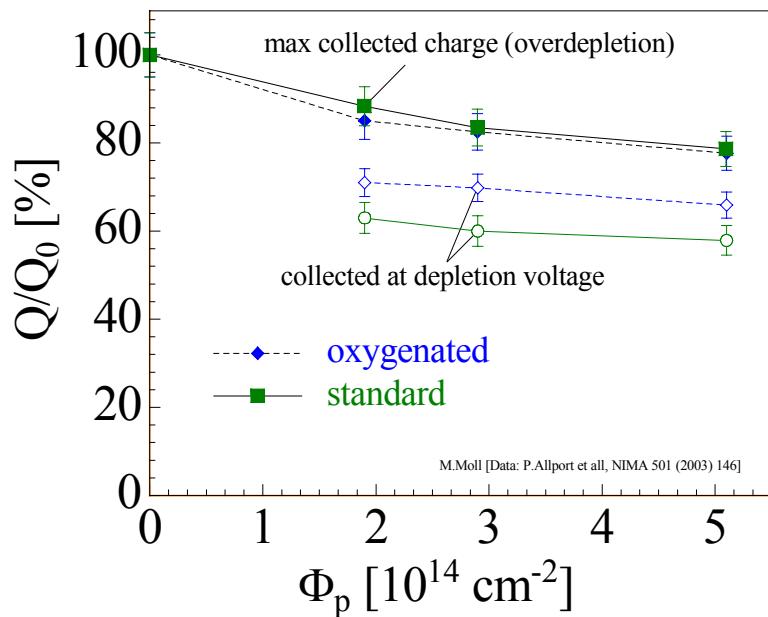
- **Two basic mechanisms reduce collectable charge:**

- trapping of electrons and holes  $\Rightarrow$  (depending on drift and shaping time !)
- under-depletion  $\Rightarrow$  (depending on detector design and geometry !)

## Example: ATLAS microstrip detectors + fast electronics (25ns)

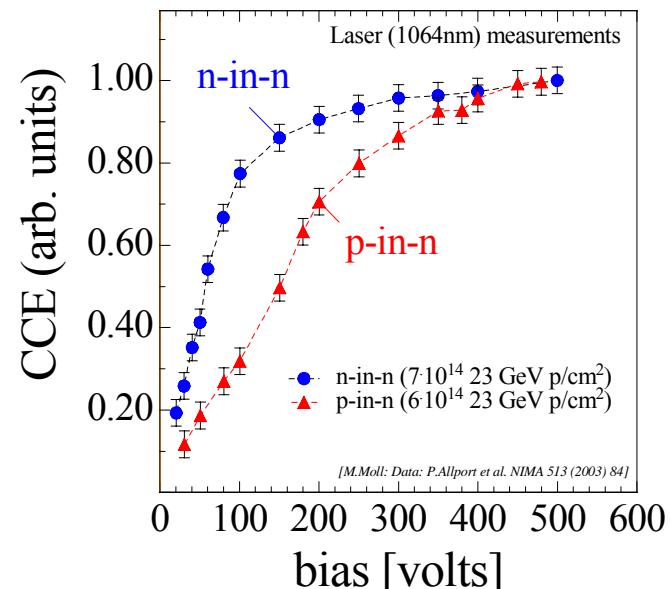
### p-in-n : oxygenated versus standard FZ

- beta source
- 20% charge loss after  $5 \times 10^{14}$  p/cm<sup>2</sup> (23 GeV)



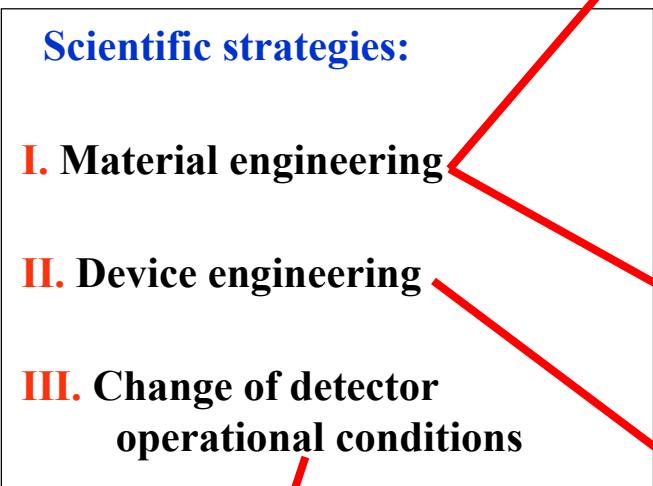
### n-in-n versus p-in-n

- same material,  $\sim$  same fluence
- over-depletion needed



Adopted from M. Moll,CERN, Bonn, Sep-05

# Approaches for Radiation Hardening



CERN-RD39  
“Cryogenic Tracking Detectors”  
„Lazarus Effect“

## Defect Engineering of Silicon

Understanding radiation damage

**Macroscopic effects and Microscopic defects**

Simulation of defect properties & kinetics

Irradiation with different particles & energies

Oxygen rich Silicon

**DOFZ, Cz, MCZ, EPI**

Oxygen dimer & hydrogen enriched Si

Pre-irradiated Si

Influence of processing technology

## New Materials

Silicon Carbide (SiC), Gallium Nitride (GaN)

Diamond: CERN RD42 Collaboration

Amorphous silicon

## Device Engineering (New Detector Designs)

p-type silicon detectors (n-in-p)

thin detectors

3D and Semi 3D detectors

Stripixels

Cost effective detectors

Simulation of highly irradiated detectors

Monolithic devices

*Adopted from M. Moll, CERN, Bonn, Sep-05*

# Different Sensor mMaterials

*Example: is SiC an option?*

Property	Diamond	GaN	4H SiC	Si
E <sub>g</sub> [eV]	5.5	3.39	3.26	1.12
E <sub>breakdown</sub> [V/cm]	10 <sup>7</sup>	4·10 <sup>6</sup>	2.2·10 <sup>6</sup>	3·10 <sup>5</sup>
μ <sub>e</sub> [cm <sup>2</sup> /Vs]	1800	1000	800	1450
μ <sub>h</sub> [cm <sup>2</sup> /Vs]	1200	30	115	450
v <sub>sat</sub> [cm/s]	2.2·10 <sup>7</sup>	-	2·10 <sup>7</sup>	0.8·10 <sup>7</sup>
Z	6	31/7	14/6	14
ε <sub>r</sub>	5.7	9.6	9.7	11.9
e-h energy [eV]	13	8.9	7.6-8.4	3.6
Density [g/cm <sup>3</sup> ]	3.515	6.15	3.22	2.33
Displacem. [eV]	43	≥15	25	13-20

Wide bandgap (3.3eV)  
 lower leakage current  
 than silicon

**Signal:**  
 Diamond 36 e/μm  
 SiC 51 e/μm  
 Si 89 e/μm

more charge than diamond

Higher displacement threshold  
 than silicon  
 radiation harder than silicon (?)

R&D on diamond detectors:  
 RD42 – Collaboration  
<http://cern.ch/rd42/>

*Result for SiC: very low CCE even for very thin devices*

Recent review: P.J.Sellin and J.Vaitkus on behalf of RD50

“New materials for radiation hard semiconductor detectors”, submitted to NIMA

*Adopted from M. Moll,CERN, Bonn, Sep-05*

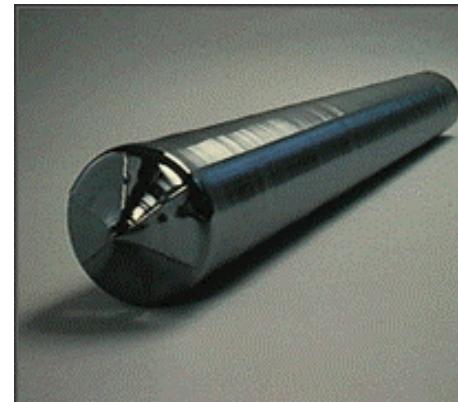
# Monocrystalline Material: Float Zone Silicon (FZ)

## ❑ Float Zone process

- Using a single Si crystal seed, melt the vertically oriented rod onto the seed using RF power and “pull” the monocrystalline ingot

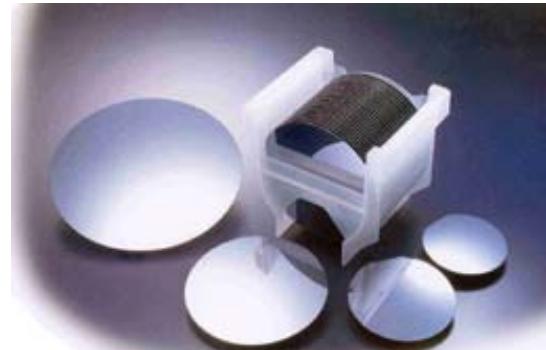


## ❑ Mono-crystalline Ingot



## ❑ Wafer production

- Slicing, lapping, etching, polishing



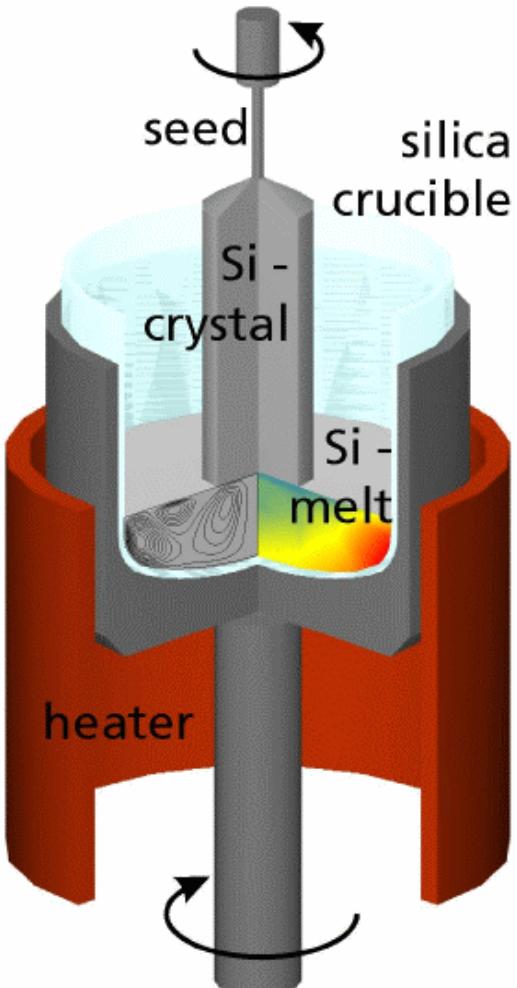
## ❑ Oxygen enrichment (DOFZ)

- Oxidation of wafer at high temperatures

*Adopted from M. Moll, CERN, Bonn, Sep-05*

# Czochralski Silicon (Cz) & Epitaxial Silicon (EPI)

## □ Czochralski silicon



- Pull Si-crystal from a Si-melt contained in a silica crucible while rotating.
- Silica crucible is dissolving oxygen into the melt  $\Rightarrow$  **high concentration of O in CZ**
- Material used by IC industry (cheap)
- Recent developments ( $\sim 2$  years) made CZ available in sufficiently high purity (resistivity) to allow for use as particle detector.

## □ Epitaxial silicon

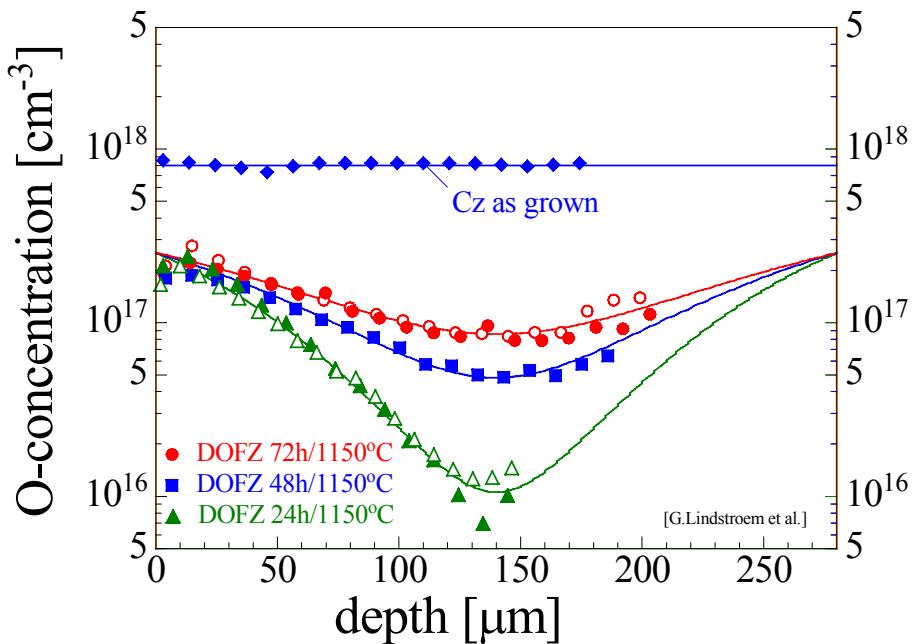
- Chemical-Vapor Deposition (CVD) of Silicon
- CZ silicon substrate used  $\Rightarrow$  **in-diffusion of oxygen**
- growth rate about  $1\mu\text{m}/\text{min}$
- excellent homogeneity of resistivity
- up to  $150 \mu\text{m}$  thick layers produced (thicker is possible)
- price depending on thickness of epi-layer but not extending  $\sim 3 \times$  price of FZ wafer

Adopted from M. Moll, CERN, Bonn, Sep-05

# Oxygen in FZ, Cz and EPI

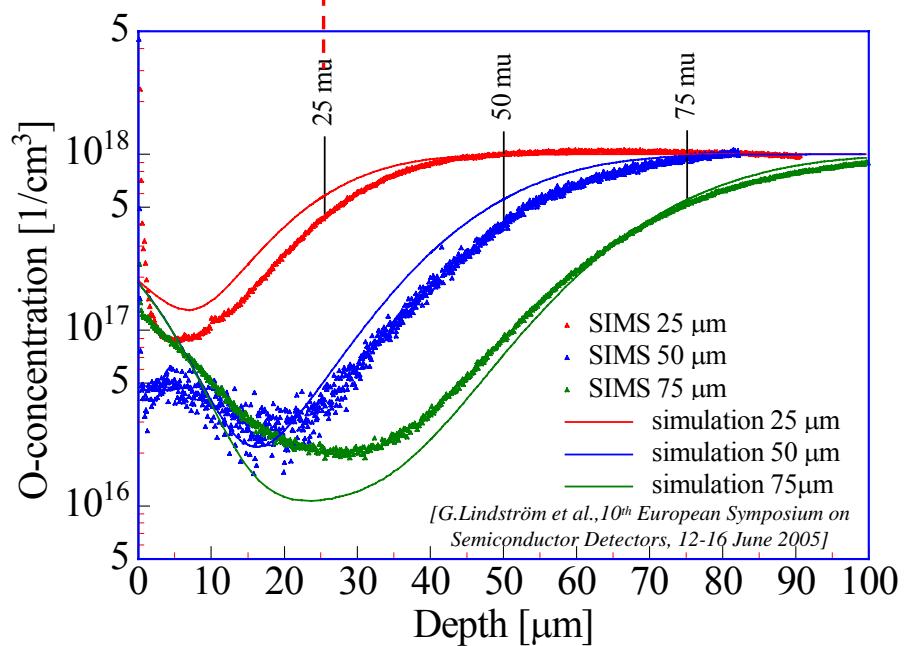
## ☐ Cz and DOFZ silicon

- CZ: high  $O_i$  (oxygen) and  $O_{2i}$  (oxygen dimer) concentration (homogeneous)
- CZ: formation of Thermal Donors possible !



- DOFZ: inhomogeneous oxygen distribution
- DOFZ: oxygen content increasing with time at high temperature

## ☐ Epitaxial silicon

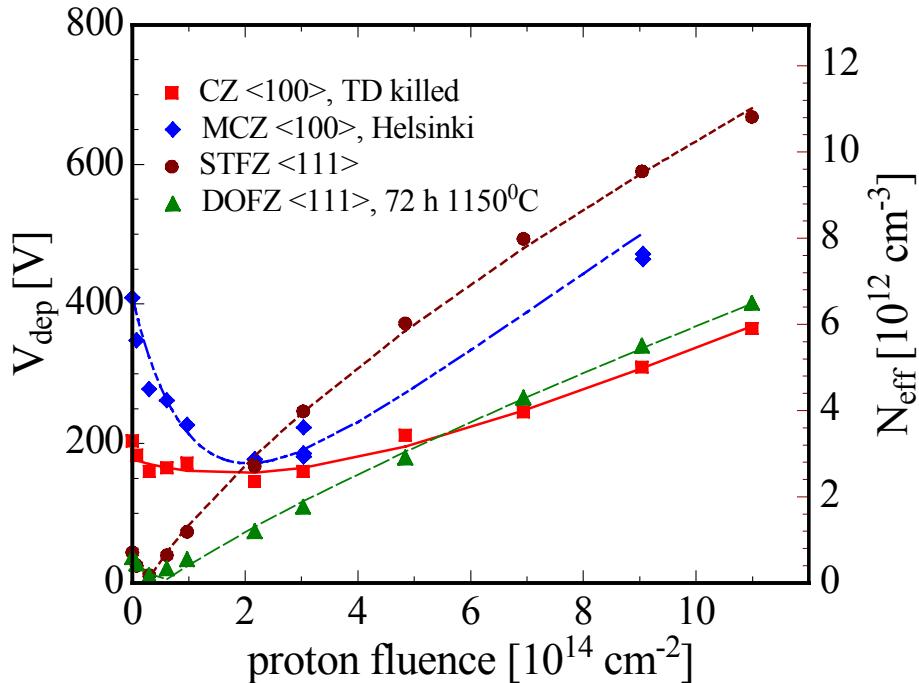


- EPI:  $O_i$  and  $O_{2i}$  (?) diffusion from substrate into epi-layer during production
- EPI: in-homogeneous oxygen distribution

# Change of Neff: FZ, DOFZ, Cz and MCz Silicon

## 24 GeV/c proton irradiation

- Standard FZ silicon
  - type inversion at  $\sim 2 \times 10^{13}$  p/cm<sup>2</sup>
  - strong N<sub>eff</sub> increase at high fluence
- Oxygenated FZ (DOFZ)
  - type inversion at  $\sim 2 \times 10^{13}$  p/cm<sup>2</sup>
  - reduced N<sub>eff</sub> increase at high fluence
- CZ silicon and MCZ silicon
  - no type inversion in the overall fluence range (verified by TCT measurements)  
(verified for CZ silicon by TCT measurements, preliminary result for MCZ silicon)  
⇒ donor generation overcompensates acceptor generation in high fluence range



- Common to all materials (after hadron irradiation):
  - reverse current increase
  - increase of trapping (electrons and holes) within  $\sim 20\%$

Adopted from M. Moll, CERN, Bonn, Sep-05

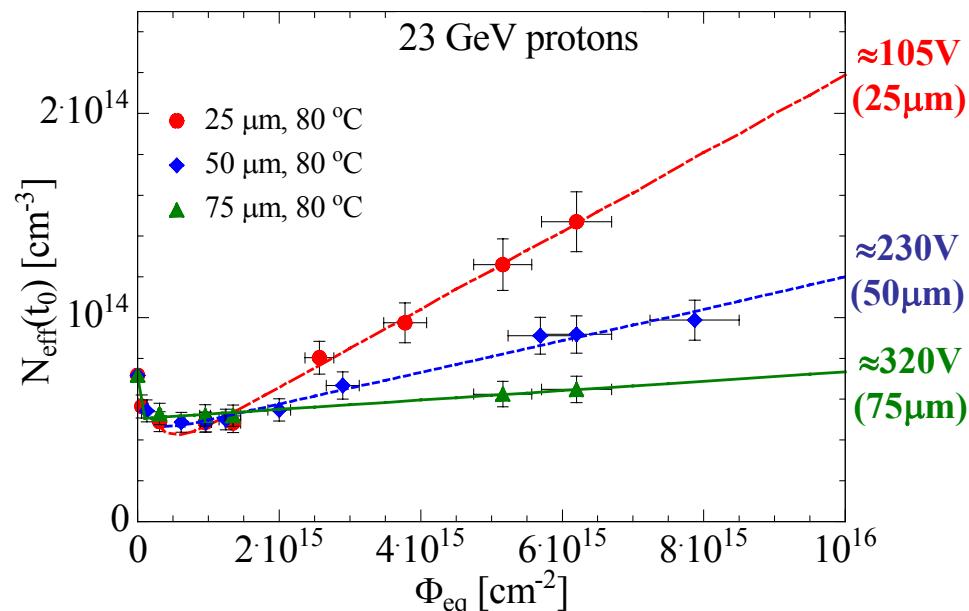
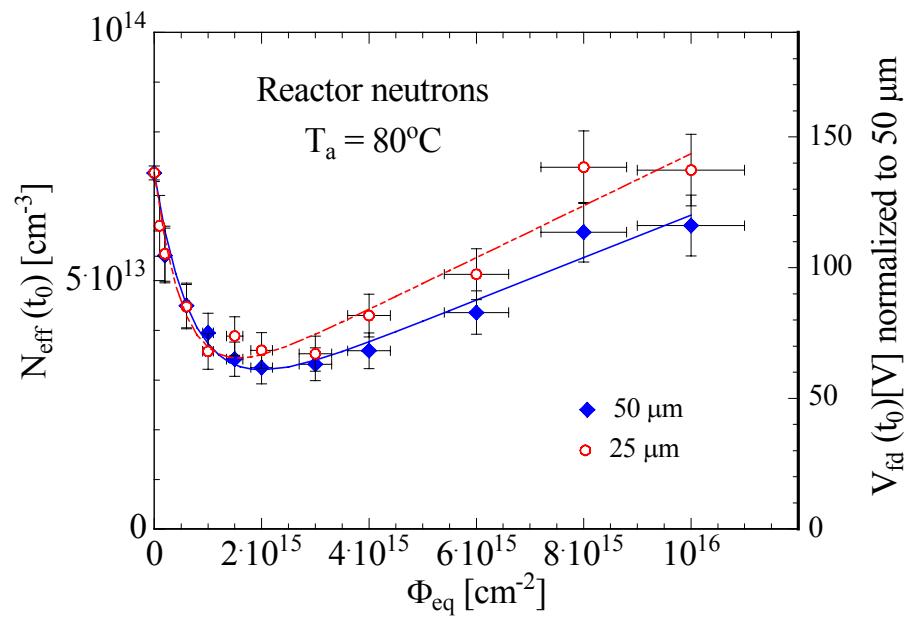
# Change of Neff: EPI Silicon

## Epitaxial silicon grown by ITME

Layer thickness: 25, 50, 75  $\mu\text{m}$ ; resistivity:  $\sim 50 \Omega\text{cm}$

Oxygen:  $[\text{O}] \approx 9 \times 10^{16} \text{ cm}^{-3}$ ; Oxygen dimers (detected via  $\text{IO}_2$ -defect formation)

G.Lindström et al., 10<sup>th</sup> European Symposium  
on Semiconductor Detectors, 12-16 June 2005



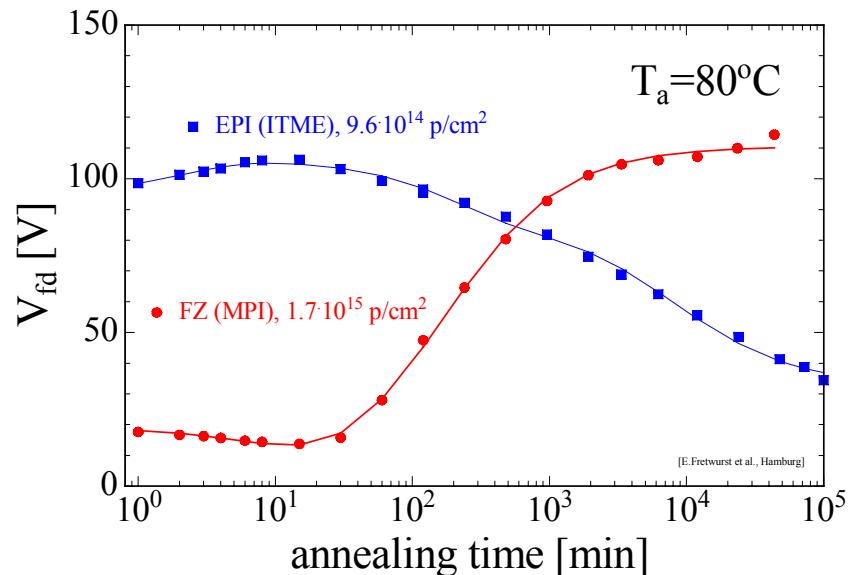
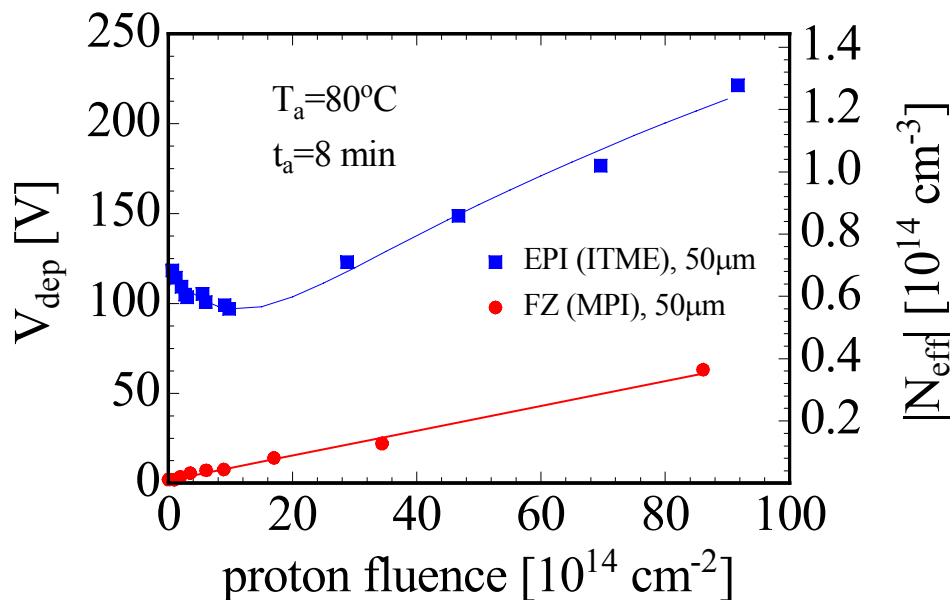
- No type inversion in the full range up to  $\sim 10^{16} \text{ p/cm}^2$  and  $\sim 10^{16} \text{ n/cm}^2$   
(type inversion only observed during long term annealing)

Proposed explanation:

introduction of shallow donors bigger than generation of deep acceptors

# Epitaxial Silicon - Annealing

- 50  $\mu\text{m}$  thick silicon detectors:
  - Epitaxial silicon (50  $\Omega\text{cm}$  on CZ substrate, ITME & CiS)
  - Thin FZ silicon (4K $\Omega\text{cm}$ , MPI Munich, wafer bonding technique)



- **Thin FZ silicon:** Type inverted, increase of depletion voltage with time
- **Epitaxial silicon:** No type inversion, decrease of depletion voltage with time  
→ No need for low temperature during maintenance of SLHC detectors!

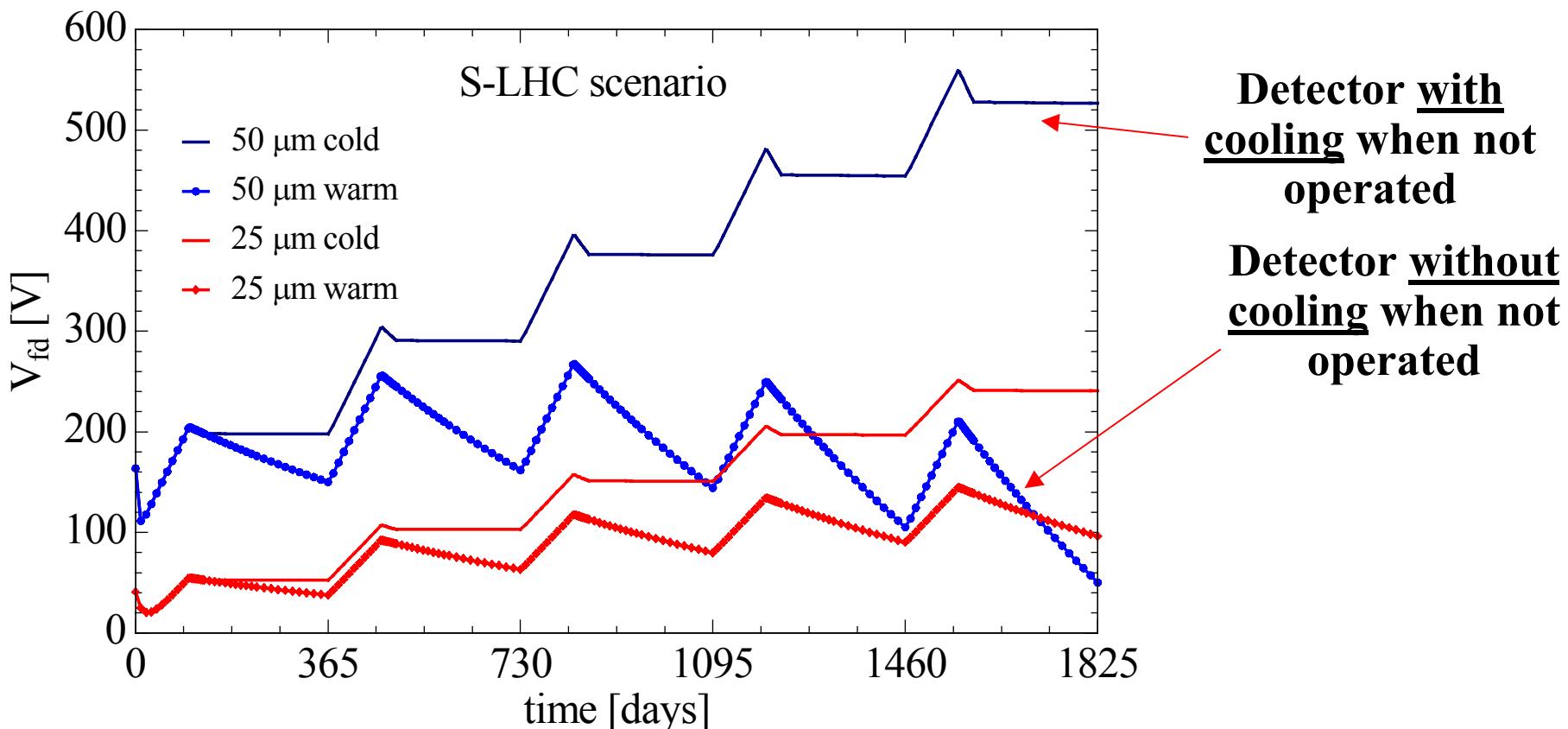
Adopted from M. Moll, CERN, Bonn, Sep-05

# Damage Projection – SLHC

## - 50 µm EPI silicon: a solution for pixels ?-

G.Lindström et al., 10<sup>th</sup> European Symposium on Semiconductor Detectors, 12-16 June 2005 (Damage projection: M.Moll)

- **Radiation level (4cm):**  $\Phi_{eq}(\text{year}) = 3.5 \times 10^{15} \text{ cm}^{-2}$
- **SLHC-scenario:** 1 year = 100 days beam (-7°C)  
30 days maintenance (20°C)  
235 days no beam (-7°C or 20°C)

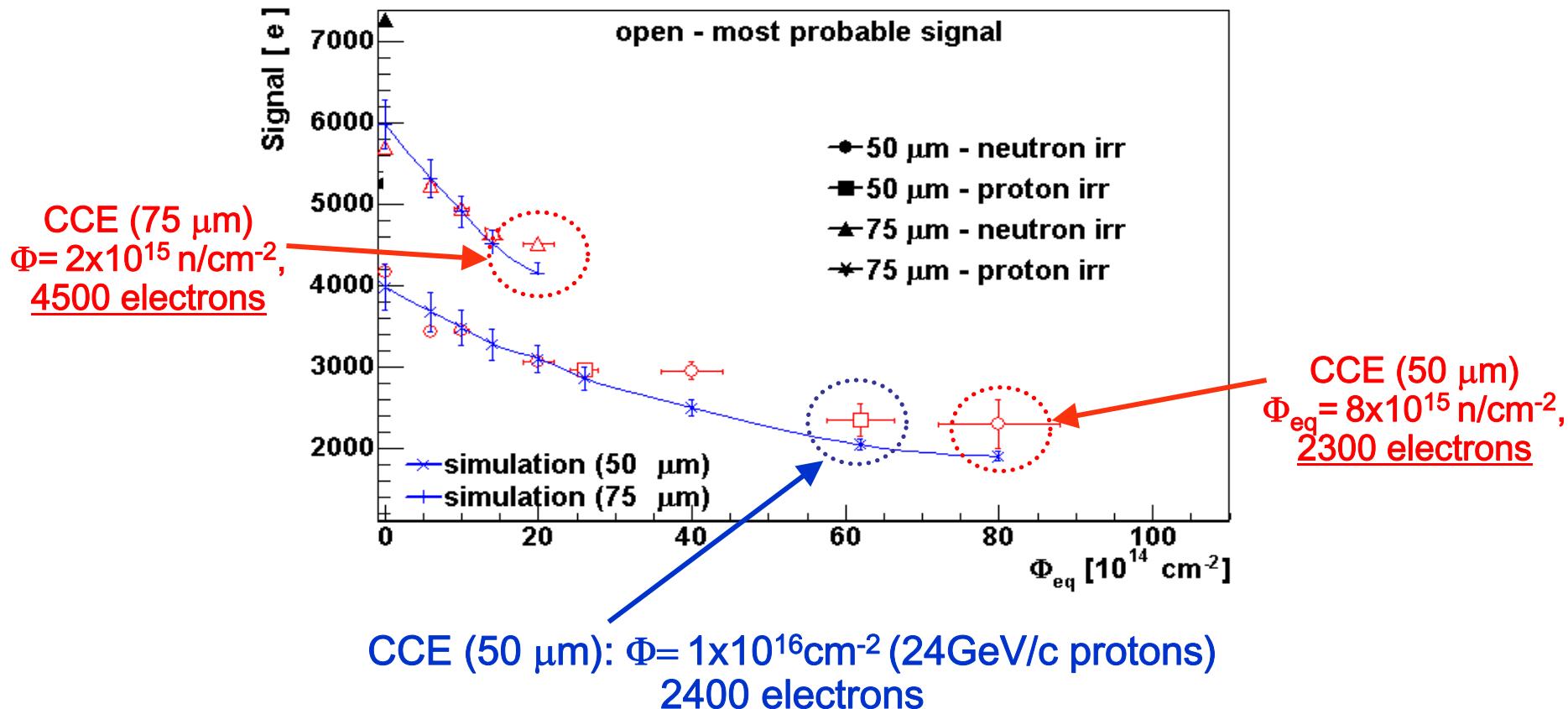


# Signal from irradiated EPI

## □ Epitaxial silicon: CCE measured with beta particles ( $^{90}\text{Sr}$ )

25ns shaping time

proton and neutron irradiations of 50  $\mu\text{m}$  and 75  $\mu\text{m}$  epi layers

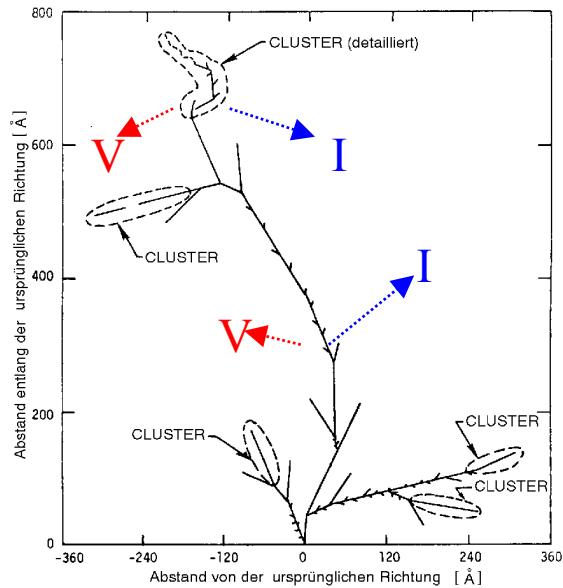
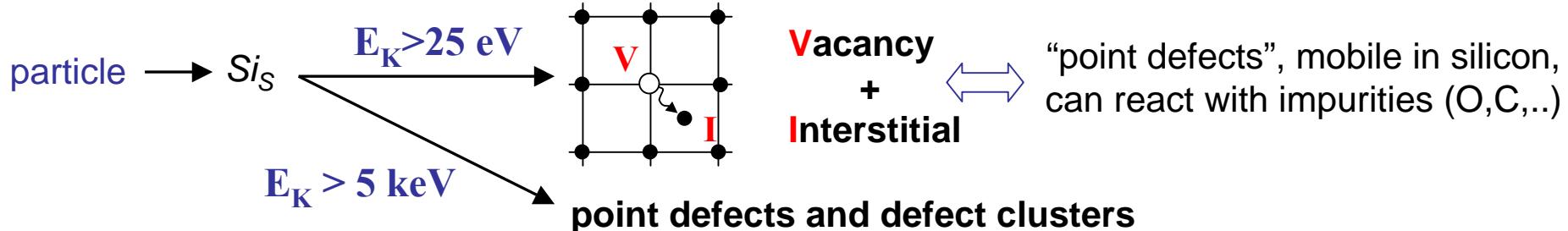


[G.Kramberger et al., RESMDD - October 2004]

Adopted from M. Moll, CERN, Bonn, Sep-05

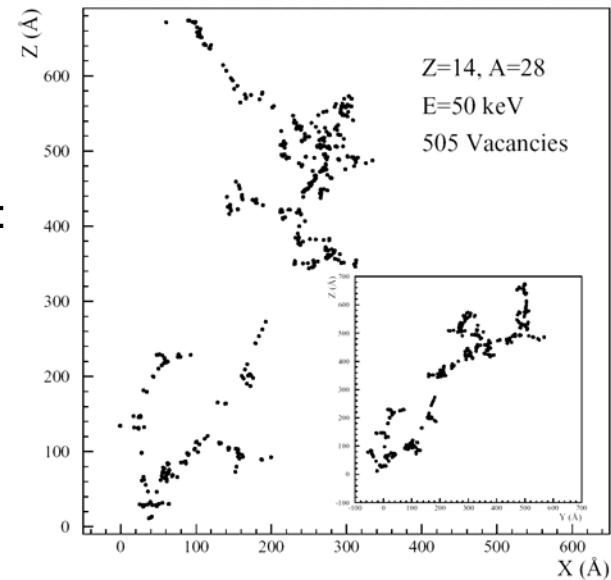
# Microscopic defects

## ■ Damage to the silicon crystal: Displacement of lattice atoms



Distribution of vacancies created by a 50 keV Si-ion in silicon (typical recoil energy for 1 MeV neutrons):

← Schematic  
[Van Lint 1980]  
Simulation →  
[M.Huhtinen 2001]



## ■ Defects can be electrically active (levels in the band gap)

- capture and release electrons and holes from conduction and valence band
- ⇒ can be charged - can be generation/recombination centers - can be trapping centers

Adopted from M. Moll, CERN, Bonn, Sep-05

# Characterization of microscopic defects

- gamma and proton irradiated silicon detectors -

## □ 2003: Major breakthrough on g-irradiated samples

For the first time macroscopic changes of the depletion voltage and leakage current can be explained by electrical properties of measured defects !

[APL, 82, 2169, March 2003]

## □ 2004: Big step in understanding the improved radiation tolerance of oxygen enriched and epitaxial silicon after proton irradiation

[I.Pintilie, RESMDD, Oct.2004]

Levels responsible for depletion voltage changes after proton irradiation:

Almost independent of oxygen content:

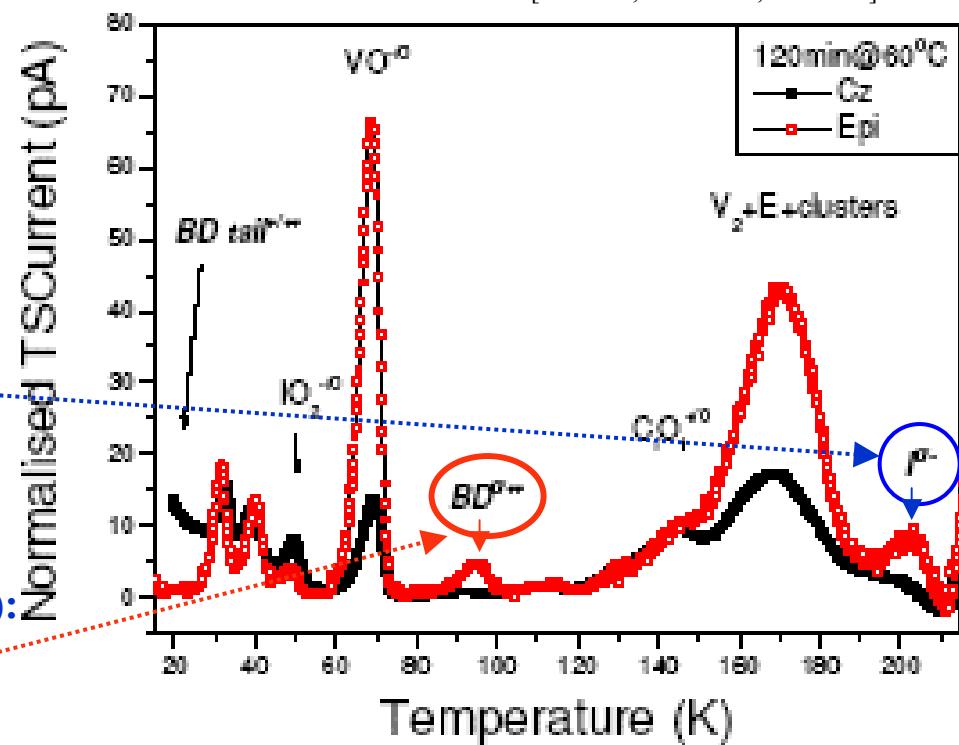
- Donor removal
- “Cluster damage”  $\Rightarrow$  negative charge

Influenced by initial oxygen content:

- I-defect: deep acceptor level at  $E_C - 0.54\text{eV}$  (good candidate for the  $\text{V}_2\text{O}$  defect)  
 $\Rightarrow$  negative charge

Influenced by initial oxygen dimer content (?):

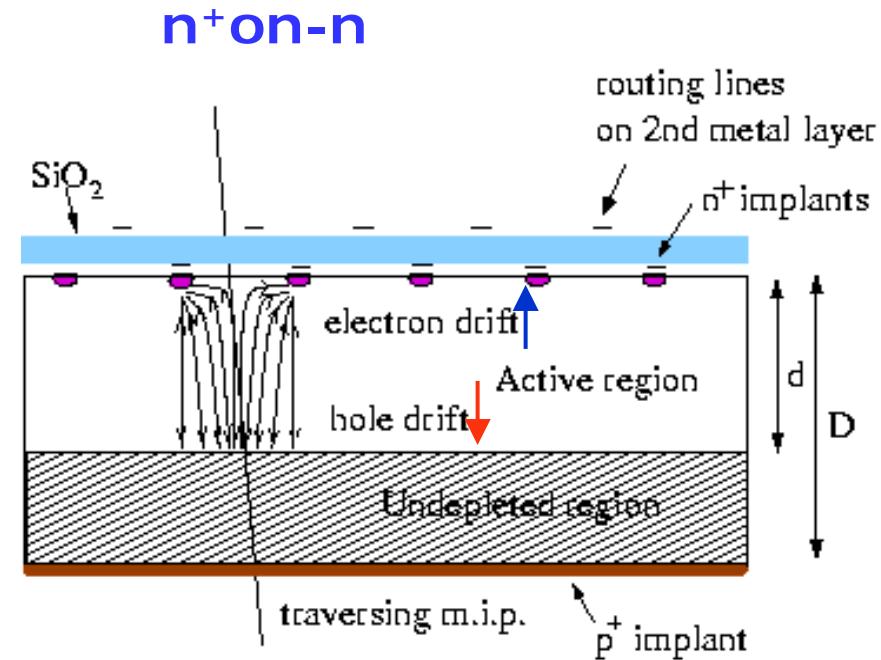
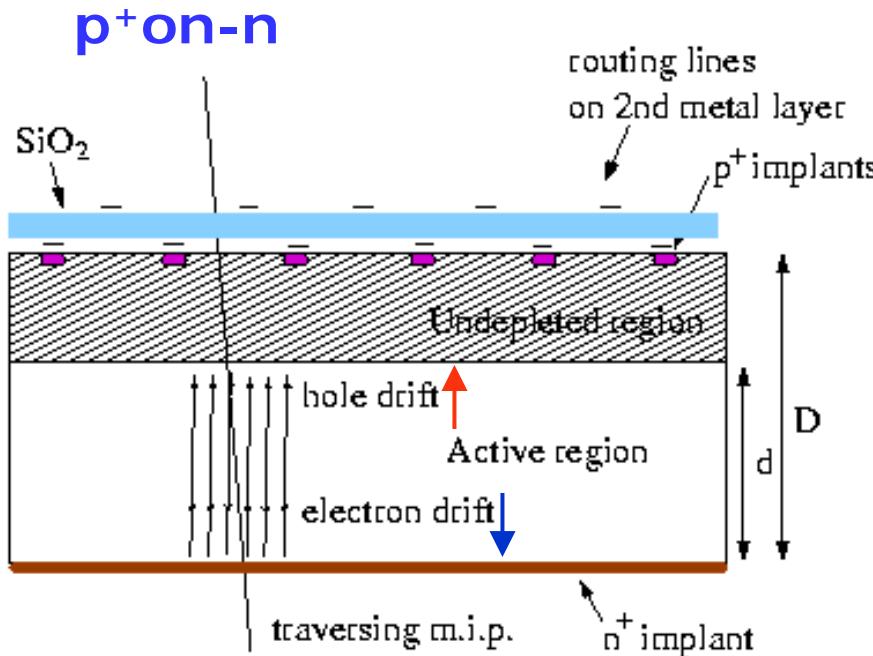
- BD-defect: bistable shallow thermal donor (formed via oxygen dimers  $\text{O}_{2i}$ )  
 $\Rightarrow$  positive charge



See further: I. Pintilie et al, NIM A 556 (2005) 56 + NIM A, in press

# Device Engineering: p-in-n versus n-in-n

n-type silicon after type inversion:



**p-on-n silicon, under-depleted:**

- Charge spread – degraded resolution
- Charge loss – reduced CCE

**n-on-n silicon, under-depleted:**

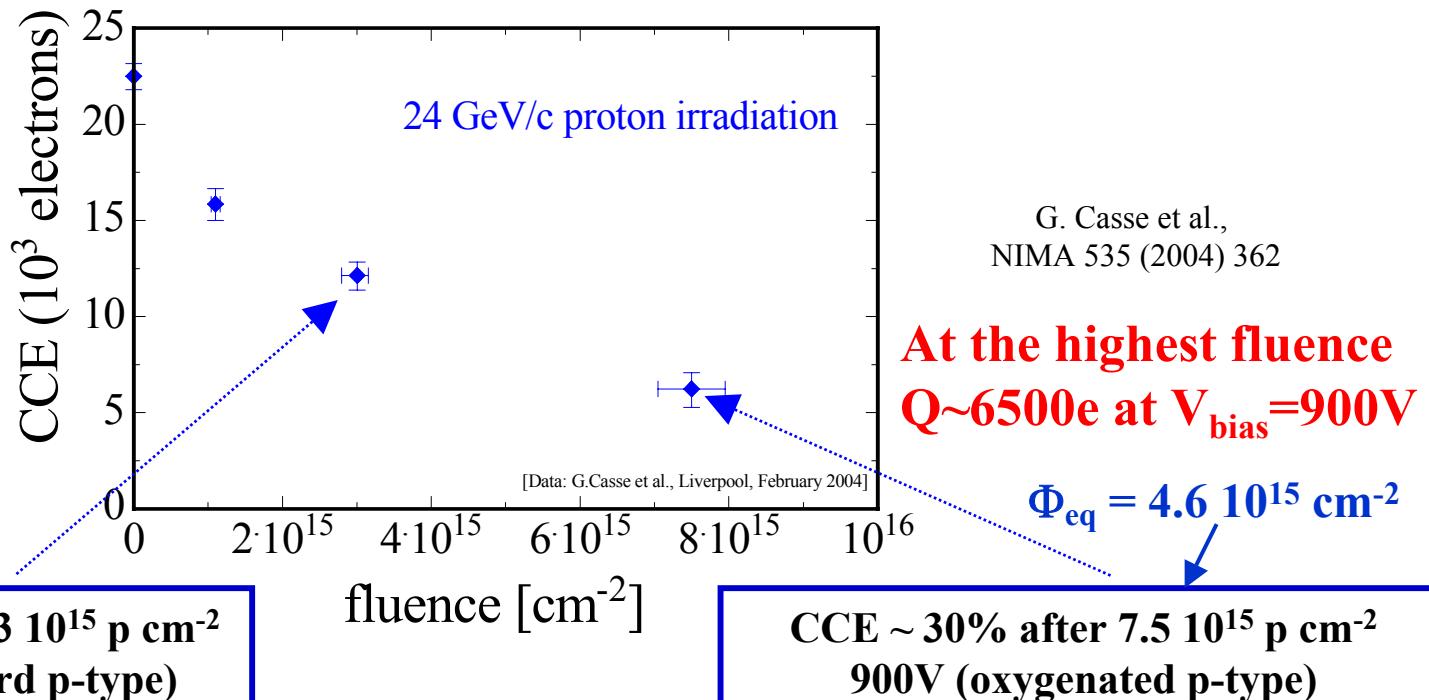
- Limited loss in CCE
- Less degradation with under-depletion
- Collect electrons (fast)

Adopted from M. Moll, CERN, Bonn, Sep-05

# n-in-p microstrip detectors

**n-in-p:** - no type inversion, high electric field stays on segmented side  
- collection of electrons

- Miniature n-in-p microstrip detectors (280mm)
- Detectors read-out with LHC speed (40MHz) chip (SCT128A)
- Material: standard p-type and oxygenated (DOFZ) p-type
- Irradiation:



Adopted from M. Moll, CERN, Bonn, Sep-05

# Does thickness pay?

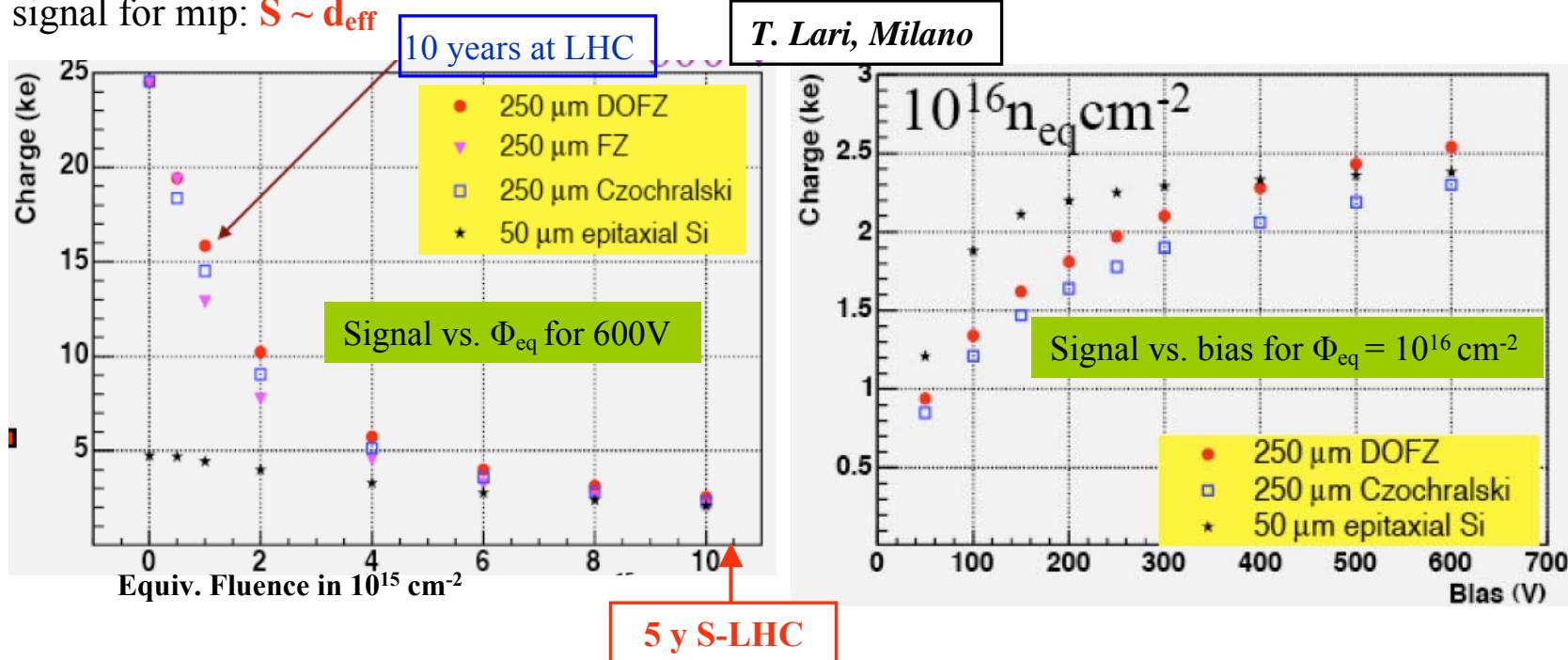
- Low fluence  $\Phi_{eq} \leq 10^{14} \text{ cm}^{-2}$ :

negligible trapping of charge carriers, signal for mip  
(minimum ionizing particle) proportional to detector thickness:  $S \sim d$

- Trapping material independent, strong increase with fluence (see above)

- Large fluence  $\Phi_{eq} \geq 5 \cdot 10^{15} \text{ cm}^{-2}$ :

signal height dominated by charge trapping, collection distance  $d_{eff} \leq 100 \mu\text{m}$   
signal for mip:  $S \sim d_{eff}$



! After  $\Phi_{eq} = 10^{16} \text{ cm}^{-2}$   $S(\text{mip}) = \text{about } 2000\text{-}2500 \text{ e}$  regardless of material and thickness !

# Summary

- At fluences up to  $10^{15}\text{cm}^{-2}$  (Outer layers of a SLHC detector) the change of depletion voltage and the large area to be covered by detectors is the major problem.
  - CZ silicon detectors could be a cost-effective radiation hard solution  
(no type inversion, use p-in-n technology)
  - p-type silicon microstrip detectors show very encouraging results:  
 $\text{CCE} \approx 6500 \text{ e}$ ;  $\Phi_{\text{eq}} = 4 \times 10^{15} \text{ cm}^{-2}$ ,  $300\mu\text{m}$ , collection of electrons,  
no reverse annealing observed in CCE measurement!
- At the fluence of  $10^{16}\text{cm}^{-2}$  (Innermost layer of a SLHC detector) the active thickness of any silicon material is significantly reduced due to trapping.  
The promising new options are:
  - Thin/EPI detectors : drawback: radiation hard electronics for low signals needed
    - e.g.  $2300\text{e}$  at  $\Phi_{\text{eq}} 8 \times 10^{15}\text{cm}^{-2}$ ,  $50\mu\text{m}$  EPI,  
..... thicker layers will be tested in 2005/2006
  - 3D detectors : drawback: technology has to be optimized  
..... steady progress within RD50
- New Materials like SiC and GaN (not shown) have been characterized .
  - CCE tests show that these materials are not radiation harder than silicon

Further information: <http://cern.ch/rd50/>

Adopted from M. Moll, CERN, Bonn, Sep-05









